

InGaN/GaN quantum wells by Molecular Beam Epitaxy at ICAP

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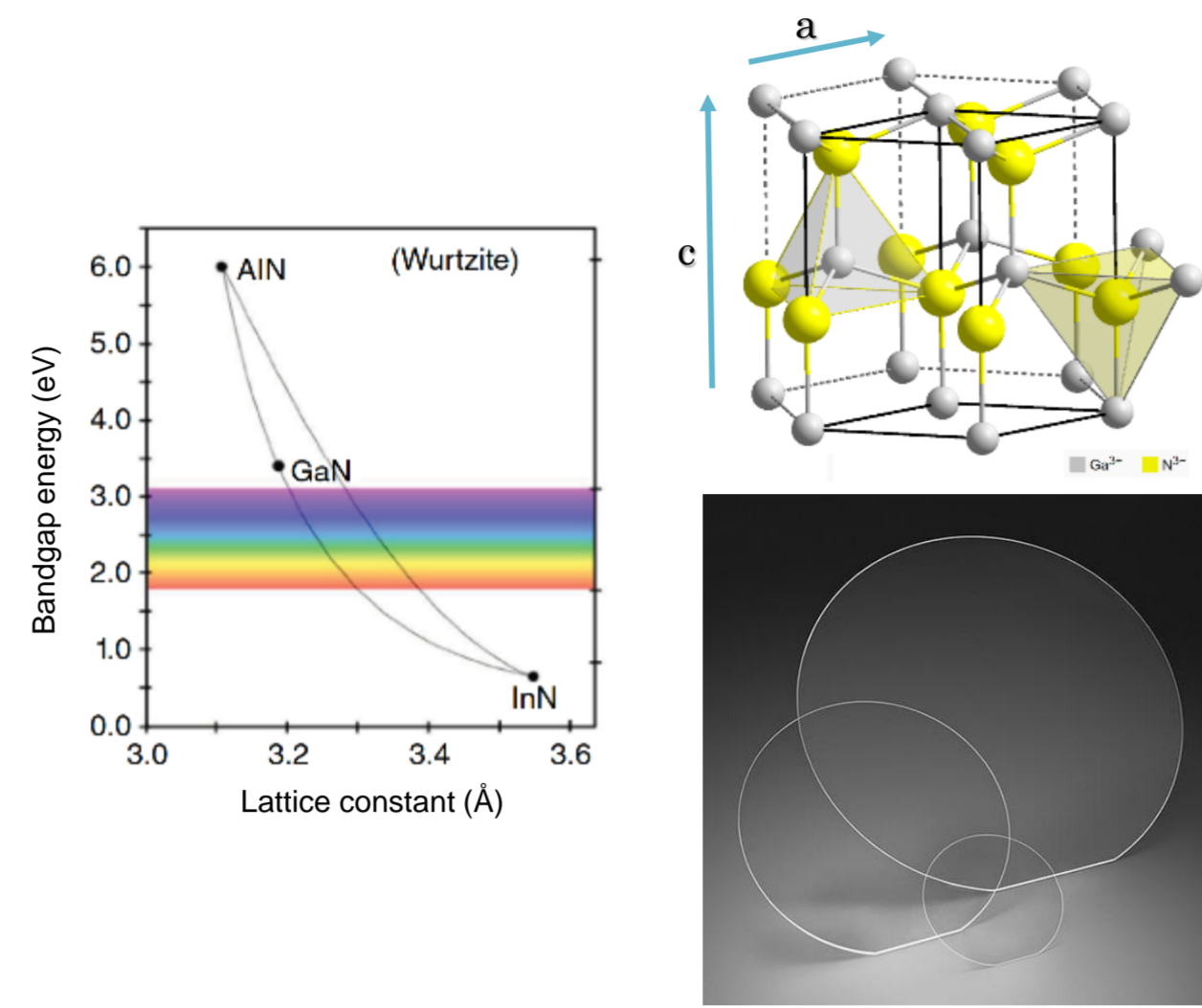
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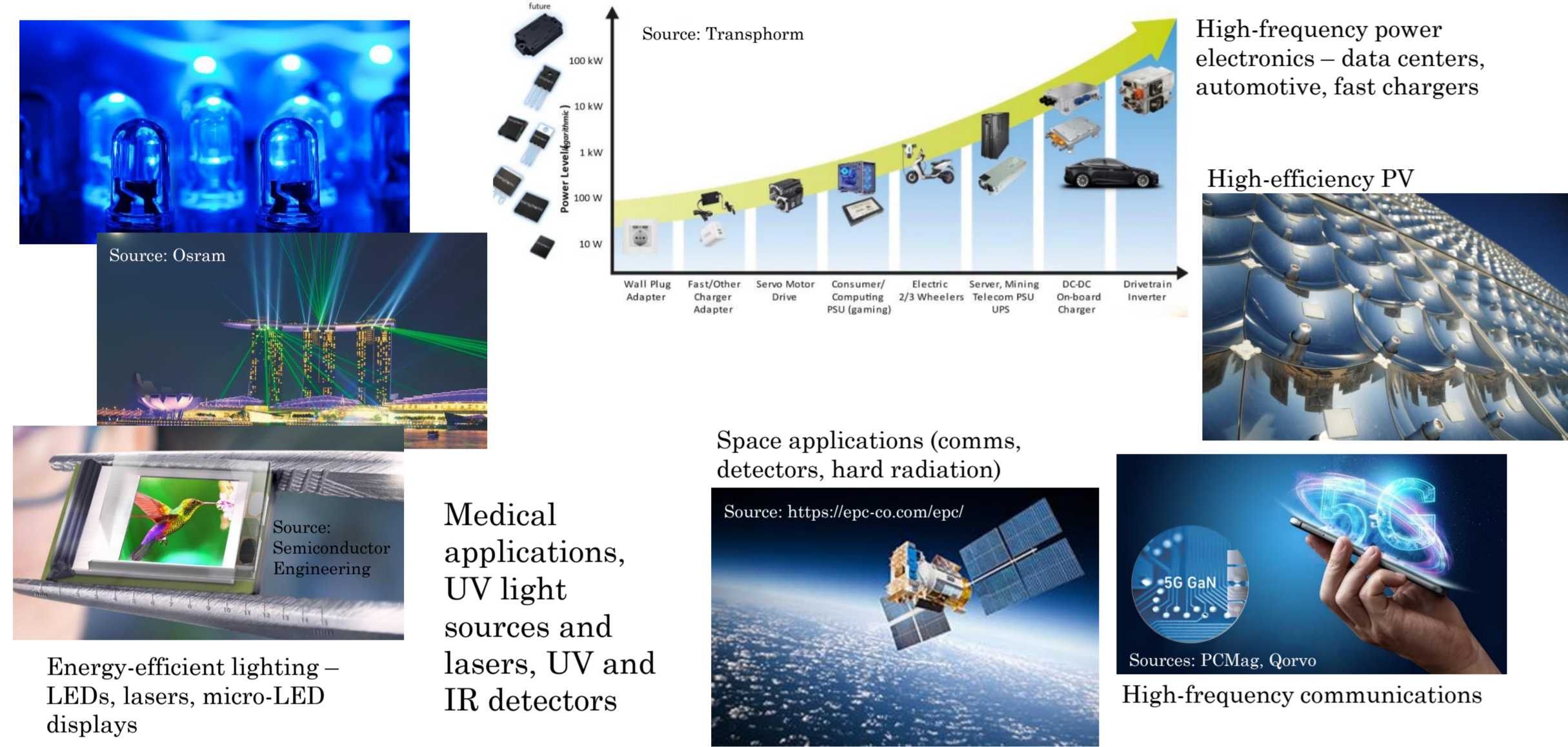
Introduction

The III-Nitride family

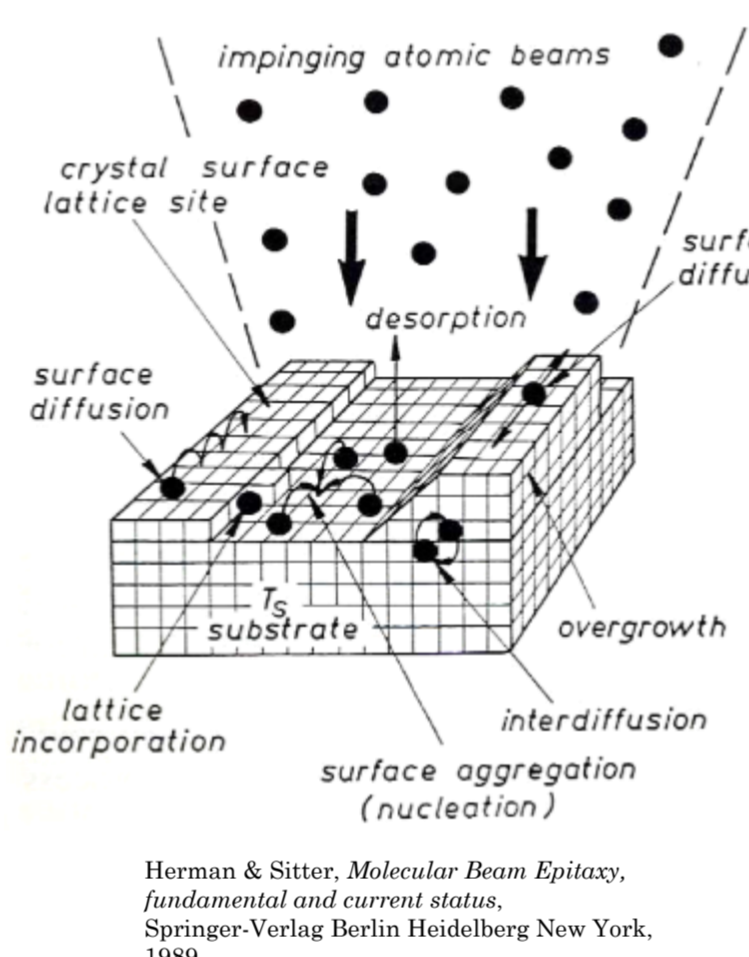
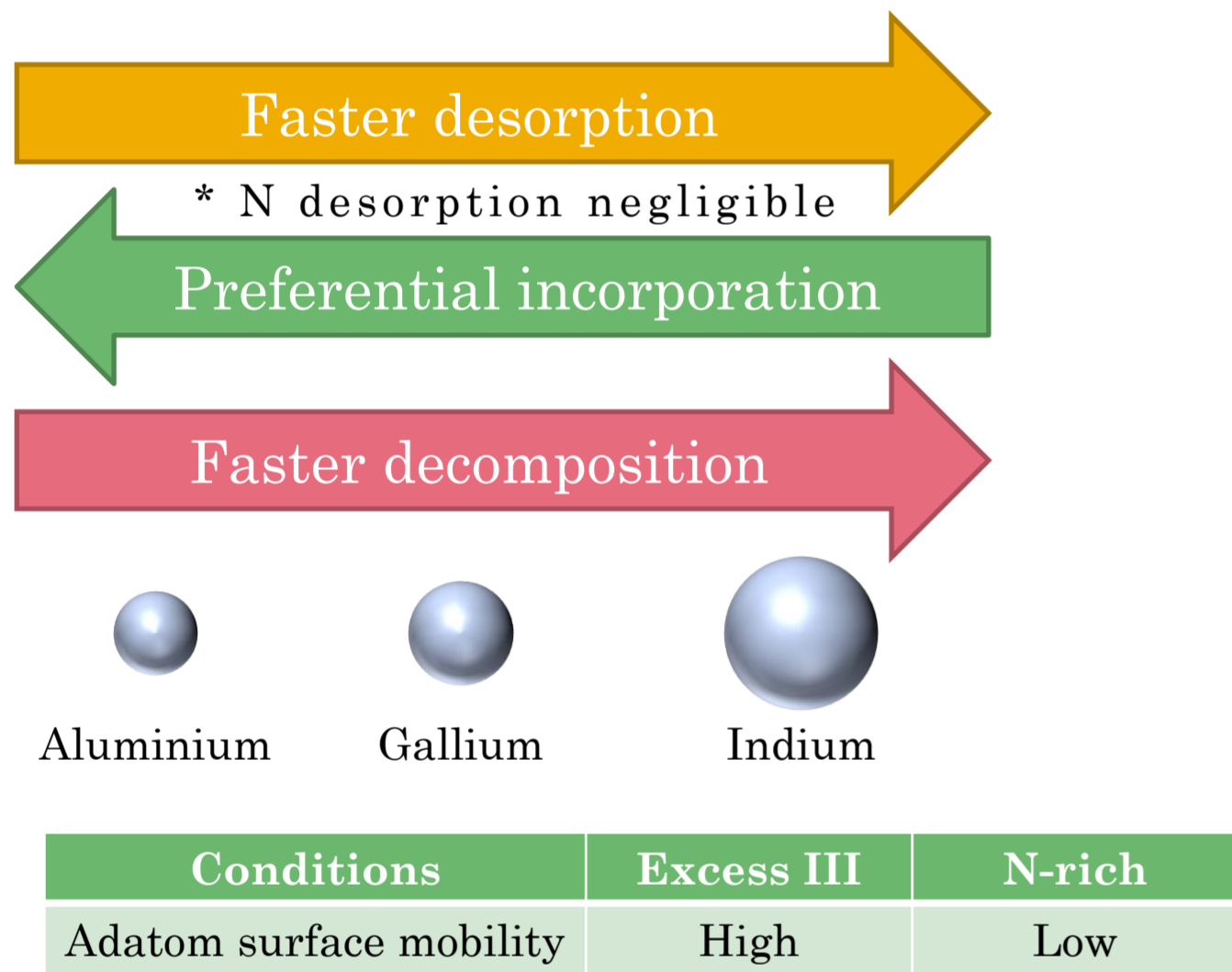
- Direct, wide bandgap semiconductors
 - High electro-optical efficiency
 - High breakdown field
- Composition-tunable properties
 - Lattice constant
 - Bandgap energy
 - Polarization
- High durability
 - Radiation hardness
 - Mechanical hardness
 - Chemical stability



III-Nitride applications

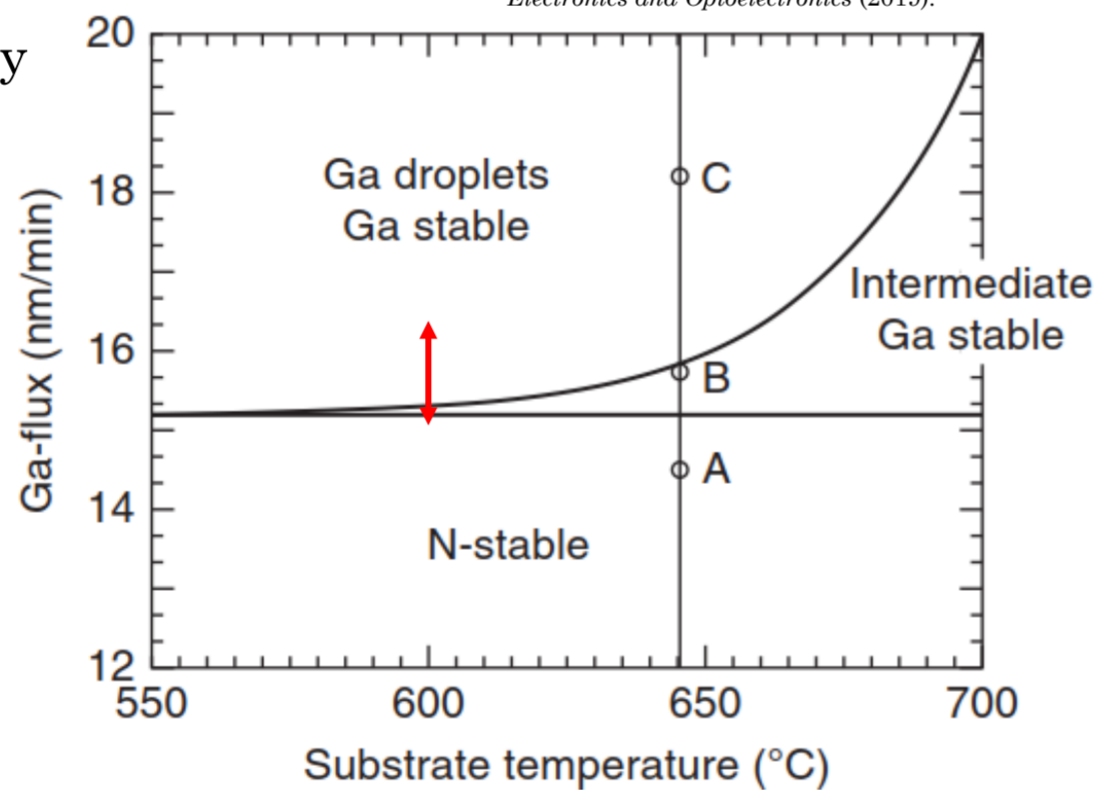


Molecular Beam Epitaxy (MBE) of III-Nitrides



Growth regimes determine crystal quality

- Challenges with InN
 - Weak In-N bond
 - InN decomposition ~450°C
 - Preferential incorporation of Ga
 - Presence of Ga stabilizes InN
 - Long In residence time at InGaN growth temperatures
 - Phase separation



- Solution: Modulate fluxes during growth
 - "Shuttle" between regimes
 - Eliminate droplets periodically
- Flux conditions:
 - $\phi(\text{Ga})/\phi(\text{N}) < 1$
 - $[\phi(\text{Ga})+\phi(\text{In})]/\phi(\text{N}) > 1$

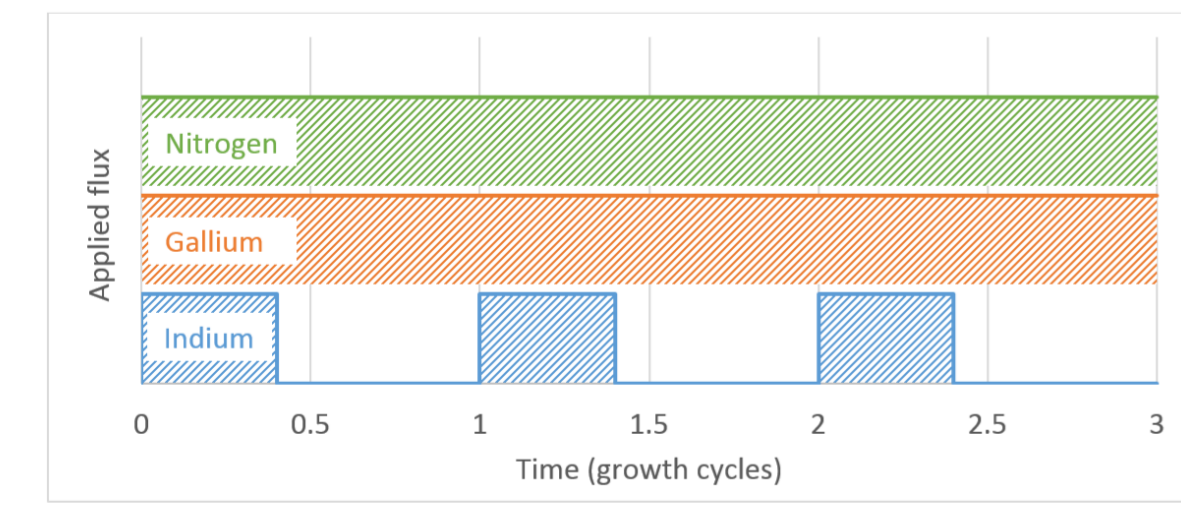
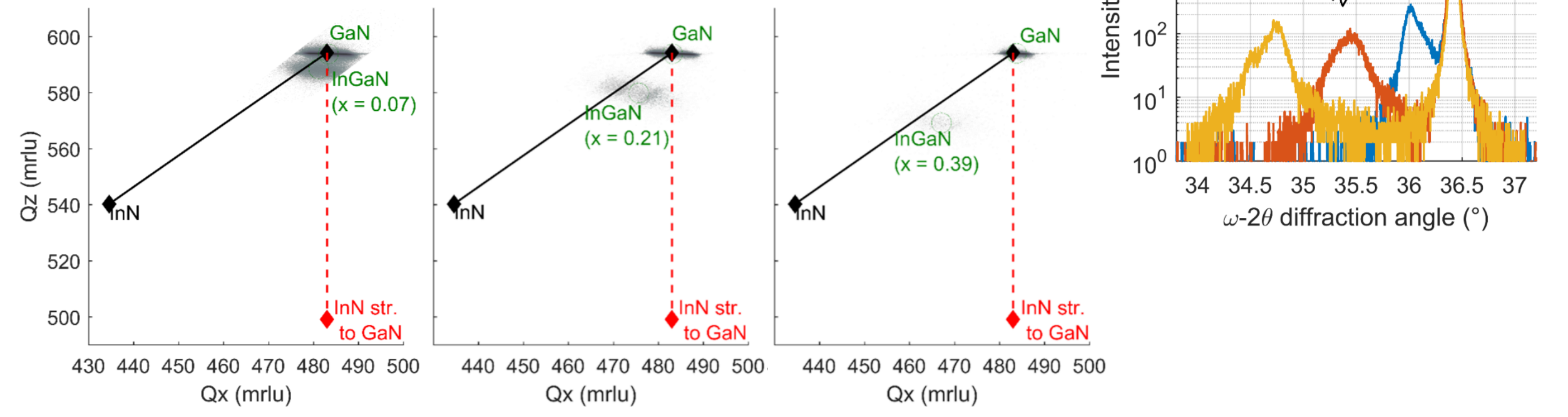


Figure 2. Flux sequence for SME InGaN (3 cycles shown). In this example the metal-rich (In+Ga/N) interval is applied for the first 40% of each cycle.

InGaN using shutter modulation

- ω -2 θ (0 0 0 4)
 - No phase separation evident in peaks
 - Relaxation leads to asymmetric peaks
- RSM (1 1 -2 4)
 - $x(\text{In}) \uparrow \Rightarrow$ Increasing relaxation
 - $x(\text{In}) \uparrow \Rightarrow$ Fainter signal

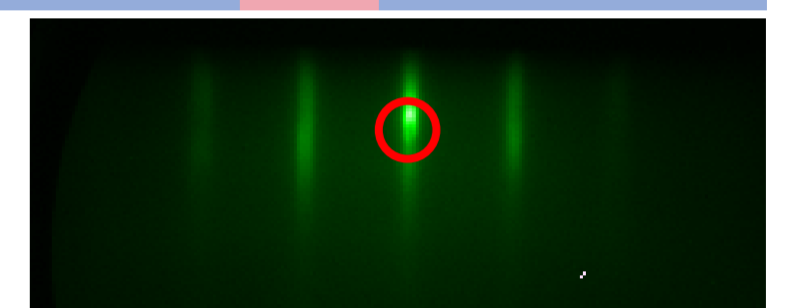


InGaN/GaN QWs using shutter modulation

- Principles of modulation:
 - $\phi(\text{Ga})/\phi(\text{N}) < 1$
 - Room for In
 - $[\phi(\text{Ga})+\phi(\text{In})]/\phi(\text{N}) > 1$
 - Excess In for metal-rich
 - InGaN QW:
 - $\text{In} > \text{In}+\text{Ga}+\text{N} > \text{Ga}+\text{N}$
 - Excess In at end
 - GaN QB: $\text{Ga} > \text{Ga}+\text{N}$
 - Ga-only time finely tuned



- Surface conditions monitored in-situ by:
 - RHEED
 - Apparent pyrometer temperature



Doping dynamics

- Preliminary metal step \rightarrow start metal-rich

- Constant volume (steady-state growth) $r = 0$:

$$[Si] = \frac{n_{Si}}{1+r\tau}$$

$$\frac{dn_{Si}}{dt} = Si_{in} - Si_{out}$$

$$\frac{d[Si]}{dt} = 0 \Rightarrow [Si]_{t>\tau} = \tau \cdot Si_{in}$$
- SME = gradually shrinking metal layer $r < 0$: slight perturbation
- Initial transient dependent on τ

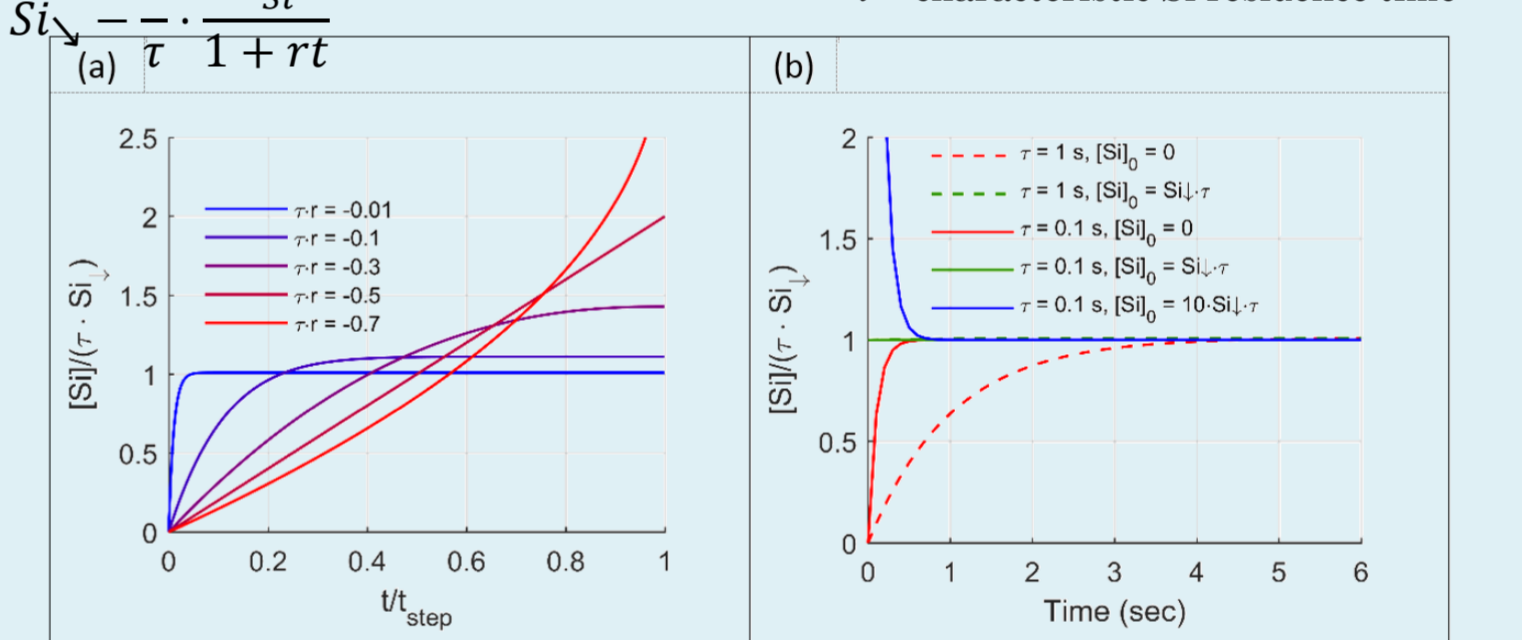
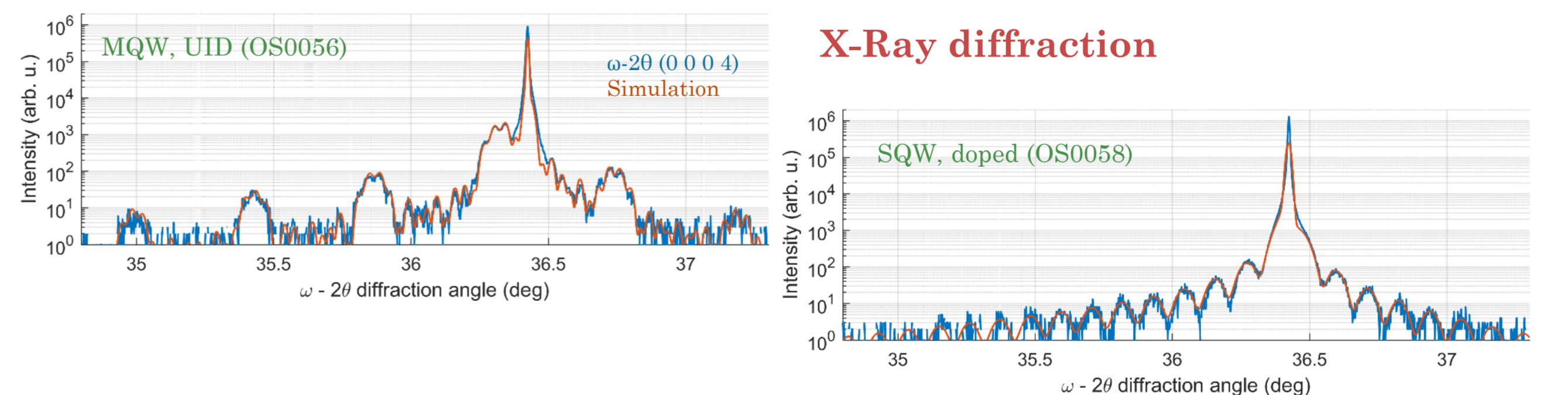
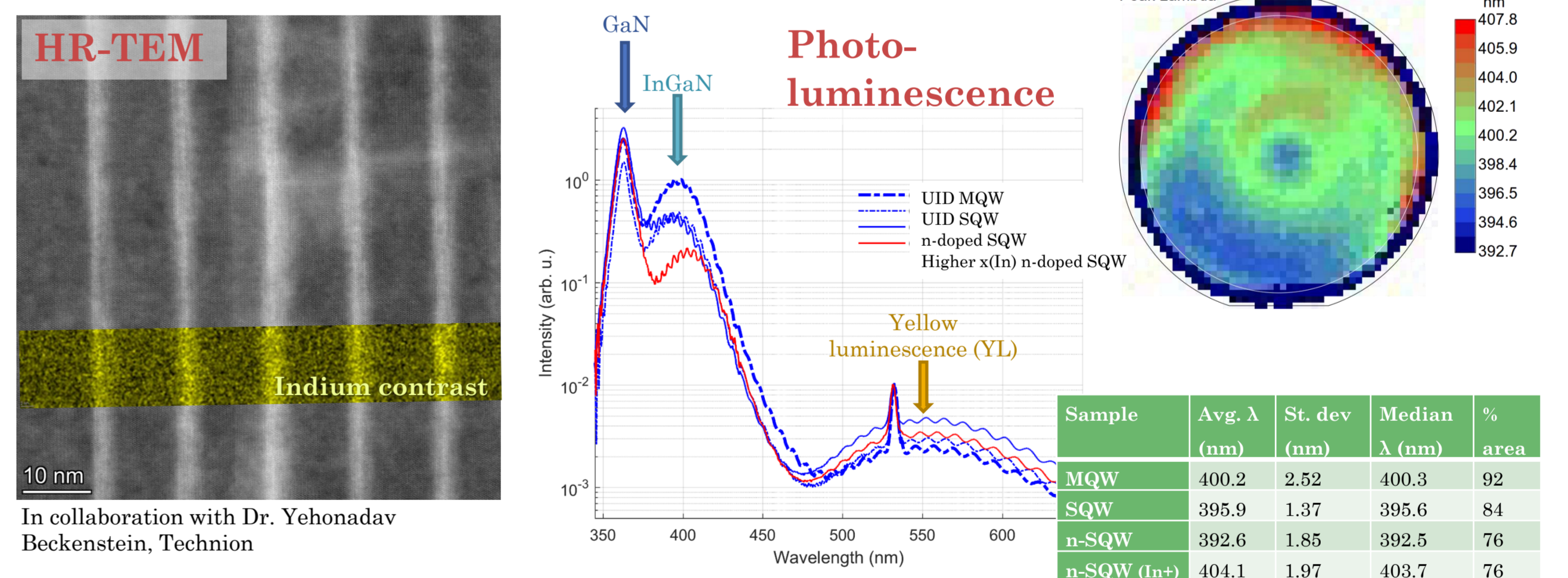


Figure A1. Calculations of Si concentration in the Ga overlayer over time, according to a variable volume Ga puddle model: (a) Development over the course of a growth step with $r = -1/\tau_{Si}$, $Si_0 = 0.1/\tau_{Si}$, $[Si]_0 = 0$, and various values of τ . (b) Initial transient in $[Si]$ at the beginning of a 95-sec growth step, at the end of which the Ga puddle is completely consumed, with $\tau = 1$ sec (dashed lines) or 0.1 sec (solid lines), and with $[Si]_0 = 0$ (red lines), $[Si]_0 = \tau \cdot Si_{in}$ (green lines) and $[Si]_0 = 10 \cdot \tau \cdot Si_{in}$ (blue line).

Characterization



Summary

- Growth of MQWs and SQWs with precise structures, sharp interfaces
- General-purpose shutter-modulated growth techniques, based on RHEED monitoring
- Doping included based on analysis, taking advantage of modulation scheme
- Repeatable, uniform PL emission but broad peaks



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Molecular Beam Epitaxy (MBE) | Metal-Organic Vapor Phase Epitaxy (MOCVD)

III-V chamber | III-N chamber | RF plasma N source (plasma-assisted MBE)

Dual growth modes: III-V / III-N

Veeco dual-chamber Gen-20A Substrates up to 100mm | Aixtron CGS dual application MOCVD Substrates up to 150mm